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Battery Degradation Modeling for Mining BEVs

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1 **Battery Degradation Overview**

2 Battery Degradation Models

3 Physics-informed Machine Learning

4 Current Challenges and Future Work



Background – What is Battery Degradation

Battery Degradation:
the decline in performance
and capacity of the battery
over time

```
graph TD; A["Battery Degradation:  
the decline in performance  
and capacity of the battery  
over time"] --> B["Degradation from Calendar Aging:  
due to battery's exposure to  
environmental conditions,  
independent of its usage"]; A --> C["Degradation from Cycling Aging:  
due to repeated charge and discharge  
cycles, directly related to how the  
battery is used"];
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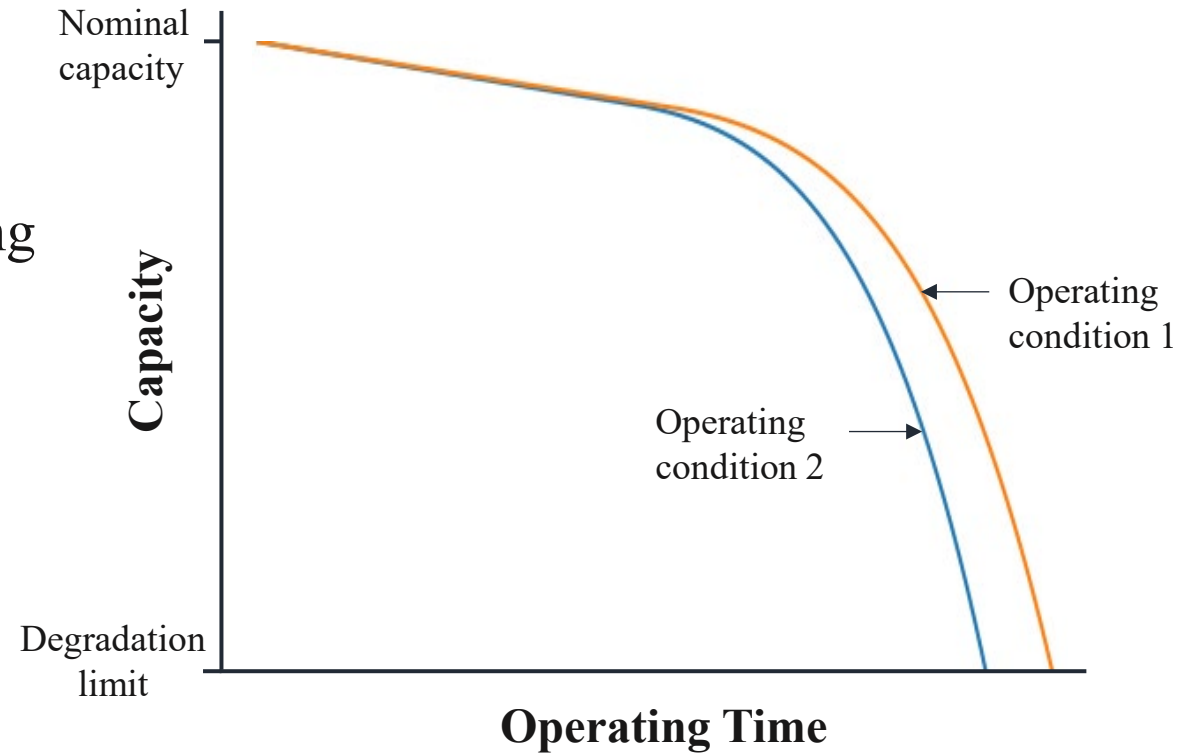
Degradation from Calendar Aging:
due to battery's exposure to
environmental conditions,
independent of its usage

Degradation from Cycling Aging:
due to repeated charge and discharge
cycles, directly related to how the
battery is used



Motivation

- Current lack of study on battery degradation for mining BEVs
- Study the different patterns of battery degradation in underground and surface mining vehicles
- Forecast battery longevity for mining duty cycles
- Investigate the factors that most significantly influence battery degradation



Battery Degradation Mechanisms

Solid–electrolyte interphase (SEI) growth

- The electrolyte reacts with the anode surface and decomposes, forming a solid layer
- Initial layer thickens during charging and discharging cause **loss of lithium inventory** from electrolyte
- Decreases capacity and increases internal resistance or impedance

Loss of active material

- Active electrode particles become electrically isolated due to cracking, mechanical stress, or binder degradation
- Reduces the amount of material available for lithium intercalation, leading to capacity loss

Lithium plating

- Li^+ ions from the electrolyte form Li metal on the graphite surface
- Plated lithium grows in needle-like, branching structures called **dendrites**, they can potentially create an **internal short circuit** if long enough
- Form extra SEI



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Physics Model - Introduction

Doyle-Fuller-Newman Model (DFN)

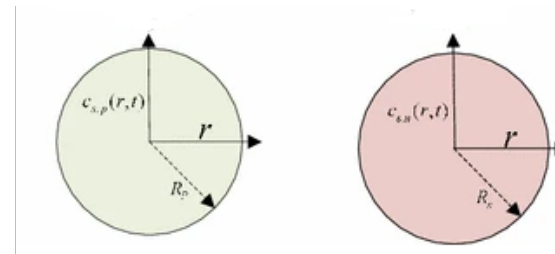
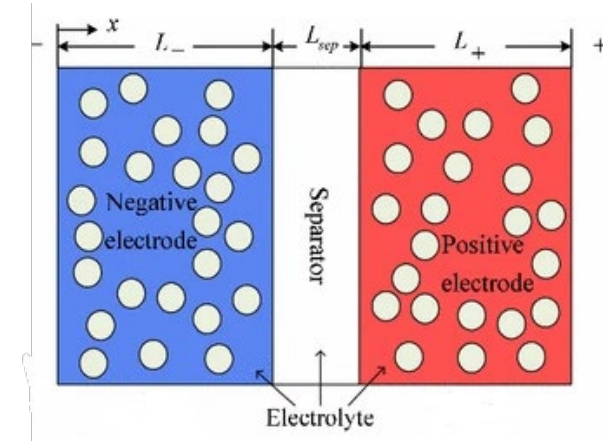
Detailed 'Pseudo-2D' model that resolves electrochemical processes across two key dimensions:

- Across the cell (x -direction): how ion concentration and potential change through the anode, separator, and cathode
- Inside particles (r -direction): lithium diffusion within the active material particles

Single Particle Model (SPM)

Highly simplified version of the DFN that assumes the electrolyte concentration and potential are spatially uniform:

- Across the Cell (x -direction): electrolyte concentration and potential are assumed uniform, and each electrode is reduced to a single representative particle
- Inside Particles (r -direction): lithium diffusion is resolved radially within the average spherical particles of the anode and cathode



Source: Kofi et al., J. Appl. Electrochem. (2017)



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Physics Model – Computational Comparison

Doyle-Fuller-Newman Model (DFN)

Single Particle Model (SPM)

Lithium Diffusion in
Solid Particles

$$\frac{\partial c_s(x, r, t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left(D_s(c_s, T) r^2 \frac{\partial c_s}{\partial r} \right)$$

$$\frac{\partial c_s(r, t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left(D_s(c_s, T) r^2 \frac{\partial c_s}{\partial r} \right)$$

Lithium Transport in
Electrolyte

$$\frac{\partial(\varepsilon_e c_e)}{\partial t} = \nabla \cdot (D_e^{\text{eff}} \nabla c_e) + \frac{1-t^+}{F} a j_{\text{tot}}(x, t)$$

Not included for SPM

SEI growth

$$\frac{\partial L_{\text{SEI}}(x, t)}{\partial t} = \frac{M_{\text{SEI}}}{\rho_{\text{SEI}} F} j_{\text{SEI}}(x, t)$$

$$\frac{dL_{\text{SEI}}(t)}{dt} = \frac{M_{\text{SEI}}}{\rho_{\text{SEI}} F} j_{\text{SEI}}(t)$$

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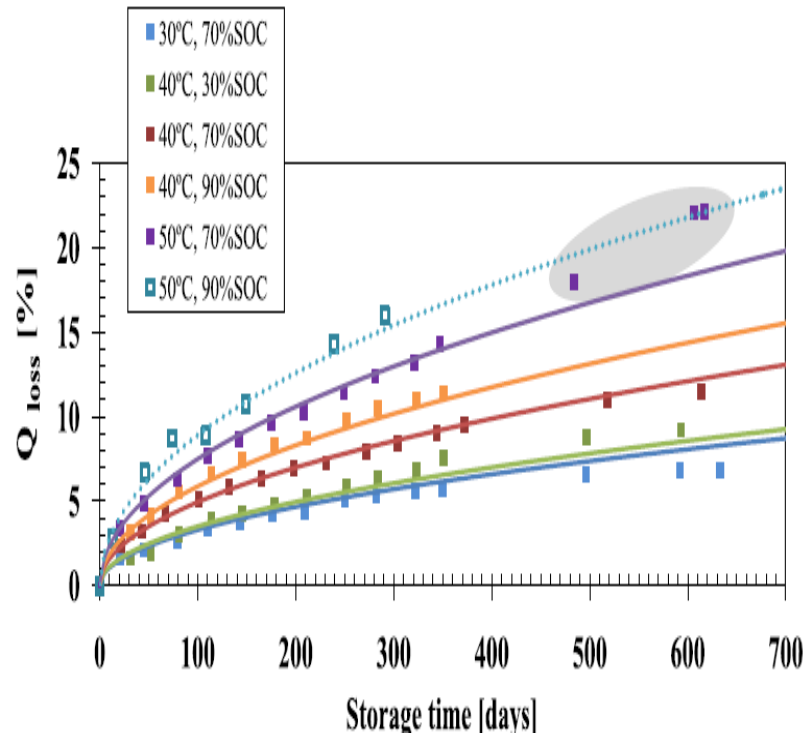
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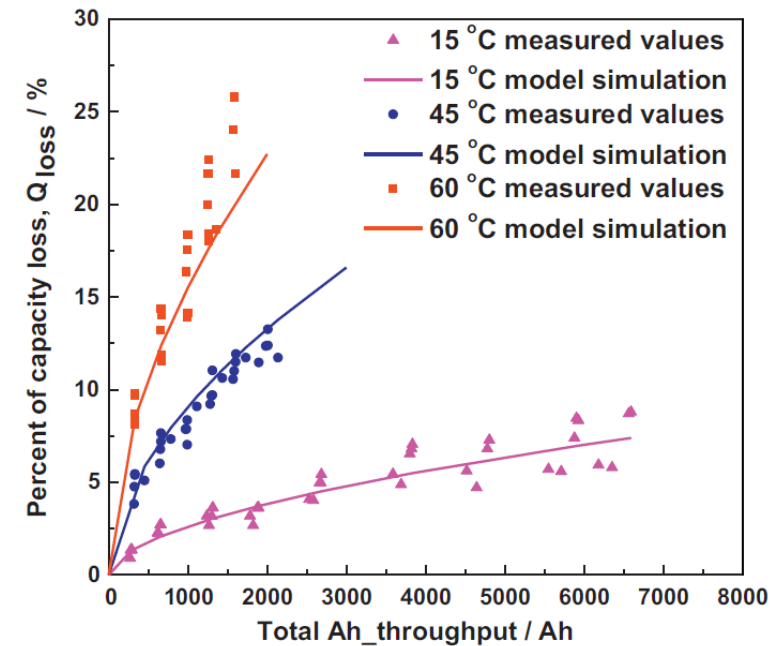
- Both DFN and SPM are modeled by solving Partial Differential Equations (PDE). PyBaMM is an open-source Python library specialized in this
- SPM is much easier to solve since it assumes uniform electrolyte concentration and has no x -dependence



Empirical Model



Source: Sarasketa-Zabala et al.(2014)

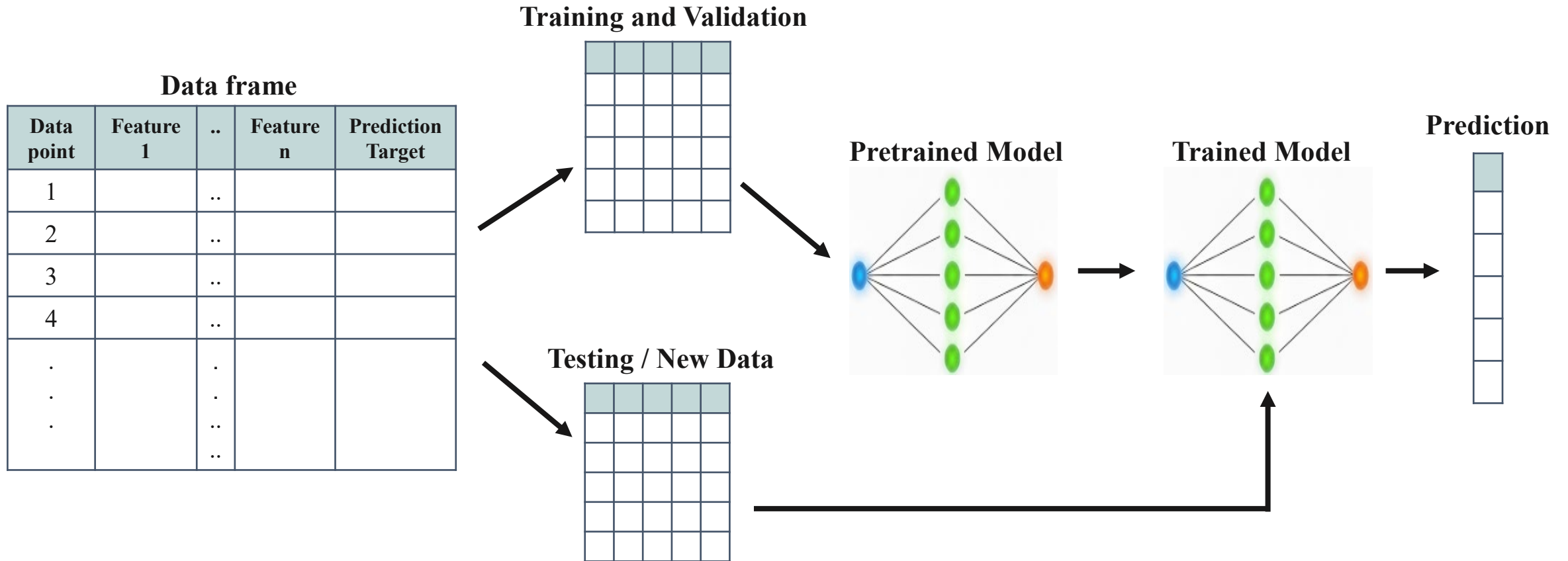


Source: Wang et al. (2011)

- For empirical models, the goal is to find a function that calculates degradation based on all parameters of interest (temperature, depth of discharge, charge/discharge rate, etc.) with minimum error using empirical data

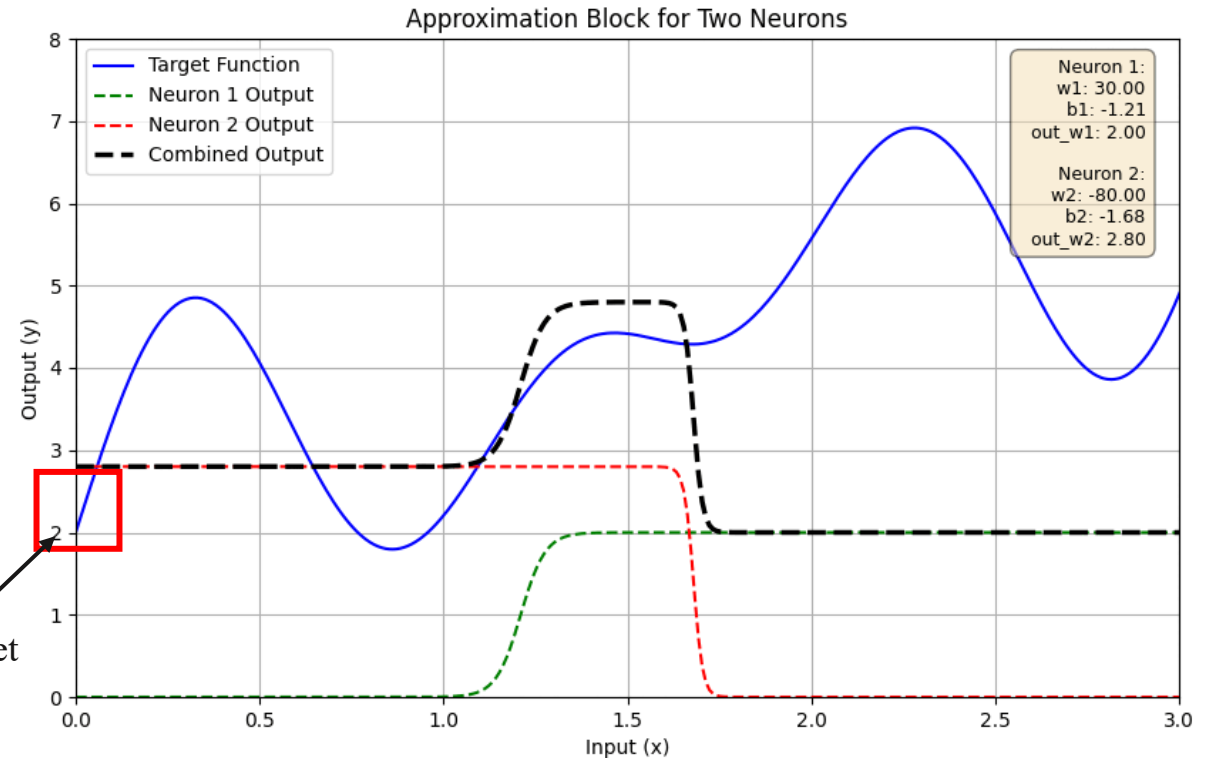
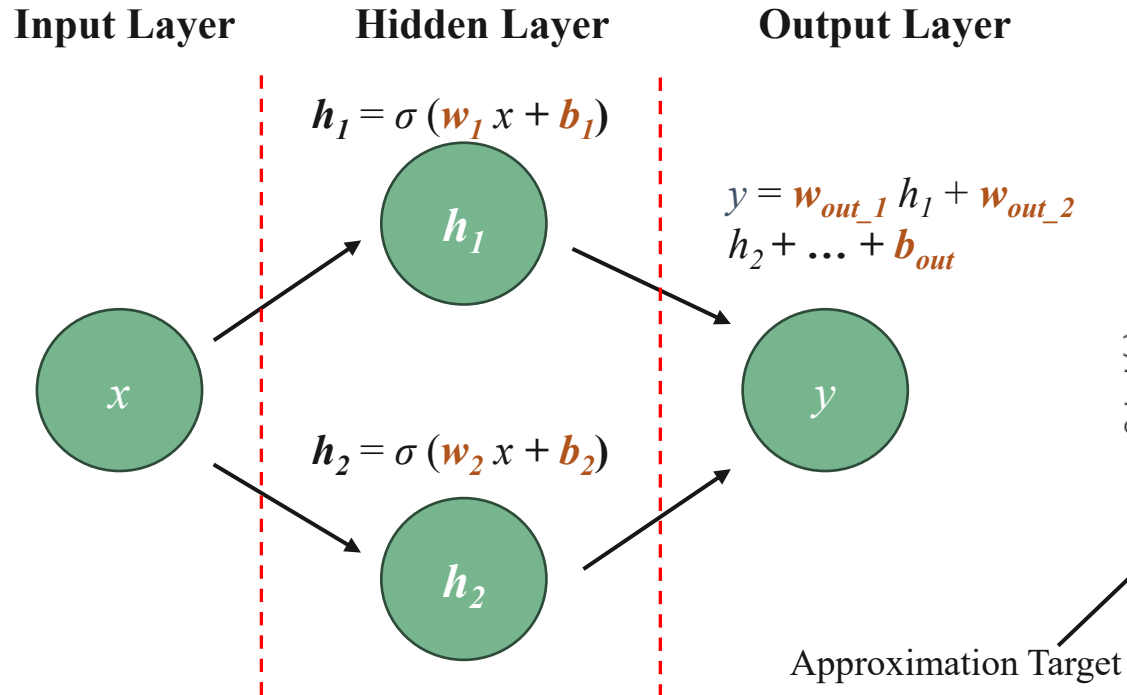


Machine Learning - Introduction



- For machine learning (ML) models, the inputs are called features, which are more flexible compared to those in physics or empirical models. ML models learn the underlying relationships between the features and the prediction target

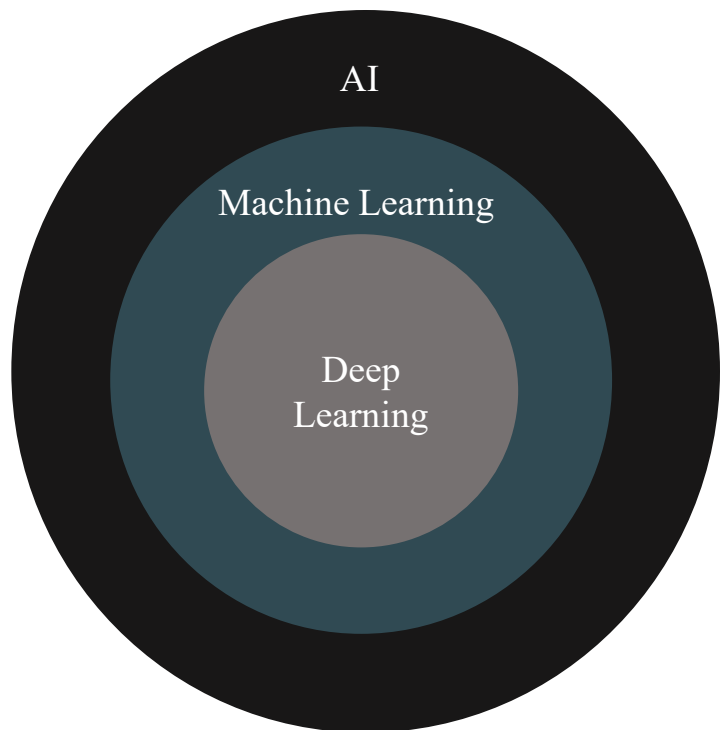
Machine Learning – Why it works



- **Universal approximation theorem:** one hidden layer, a non-linear activation function, and enough neurons can approximate any continuous function to a certain degree



Machine Learning – AI and Machine Learning



Multilayer Perceptron (MLP)

- basic deep learning model
- number prediction
- label classification

Convolutional Neural Network (CNN)

- image classification
- object identification and detection

Generative AI

- content creation

Agentic AI

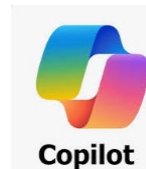
- automated project management
- operate independently without constant human input



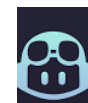
ChatGPT



Gemini



Copilot



GitHub
Copilot



Agent Builder



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Parameters for Models

Model type	Input parameters	Field obtainable
Physics (SPM)	Electrode capacity	✗
	Average particle radius	✗
	Reaction kinetics (exchange current)	✗
	Solid diffusion coefficients	✗
	SEI growth / side reaction rates	✗
	Operating data (current, voltage, temperature, SoC)	✓
Empirical	Equivalent full cycles (EFC)	✓
	Operating data (current, voltage, temperature, SoC)	✓
	Calendar aging usage (SoC storage, Temperature)	✓
Machine Learning	Flexible, can be field data or electrochemical properties or both	✓



Model Comparison

Model	Pros	Cons
Physics	<ul style="list-style-type: none">• Can be used with limited field data• Provides physical interpretability and allows for extrapolation	<ul style="list-style-type: none">• Requires electrochemical properties that are unobtainable from field data
Empirical	<ul style="list-style-type: none">• Very fast to compute• Good for trend prediction• Interpretable	<ul style="list-style-type: none">• Only reliable for the operating conditions and data used for fitting• Little physical insight• Requires full-life data from multiple batteries
Machine Learning	<ul style="list-style-type: none">• Very flexible about input features• Can capture hidden relationships between input features and prediction target• Good at handling complex non-linearity	<ul style="list-style-type: none">• Poor extrapolation outside training distribution• Limited interpretability• Requires full-life data from multiple batteries



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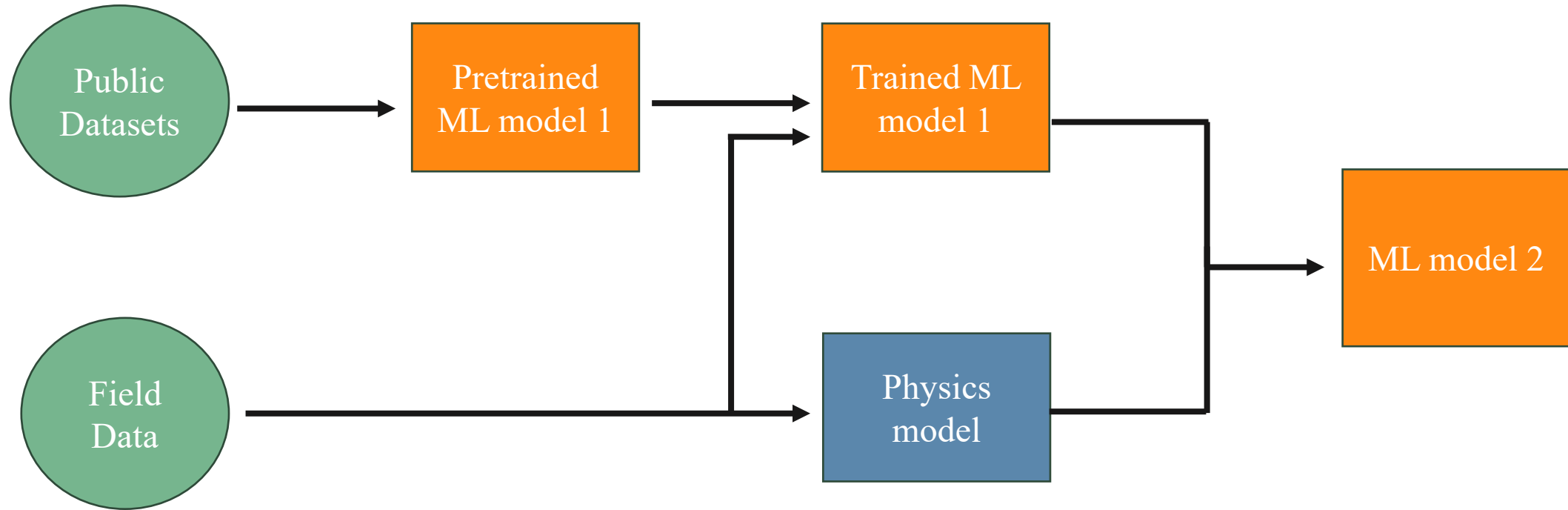
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Physics-informed Machine Learning



- In the proposed Physics-informed Machine Learning model, public datasets are first used to train a ML model to capture the relationship between cycling data and capacity. Field data are then passed through both the trained ML model and a physics model. The physics states and the residuals between the two models are used as input features to train a final ML model
- It offers greater interpretability than a standalone ML model and mitigates the limited field data challenge for ML

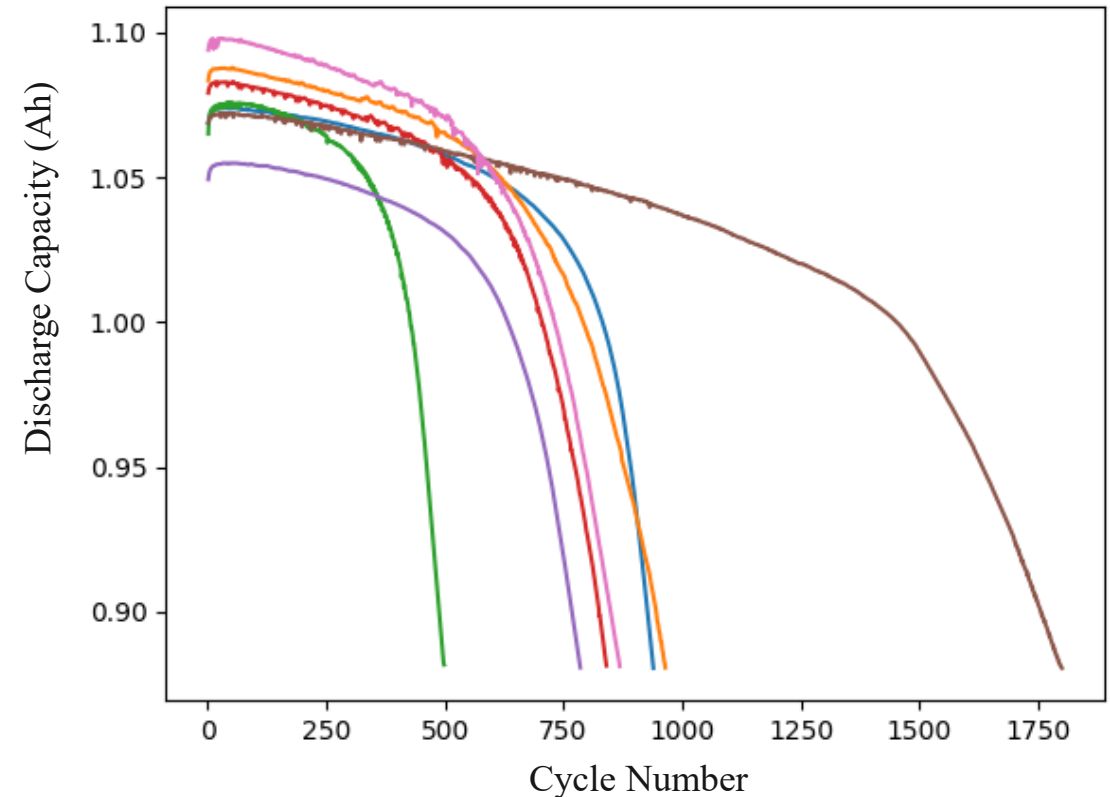


ML Model 1 – Data Overview

	Chemistry	Number of Cells	Nominal Capacity (Ah)	Temperature (°C)
1	LFP	200	1.1	30
2	NMC	55	2	~ 20 (room)
3	NCA	66	3.5	25, 35, 45
4	NMC	55	3.5	25, 35, 45
5	NMC + NCA	9	2.5	25

Datasets: <https://data.mtr.io/1/projects/5c48dd2bc625d700019f3204>
<https://zenodo.org/records/10963339>
<https://zenodo.org/records/6405084>

Example: Discharge capacity vs cycle number for 8 cells from the dataset

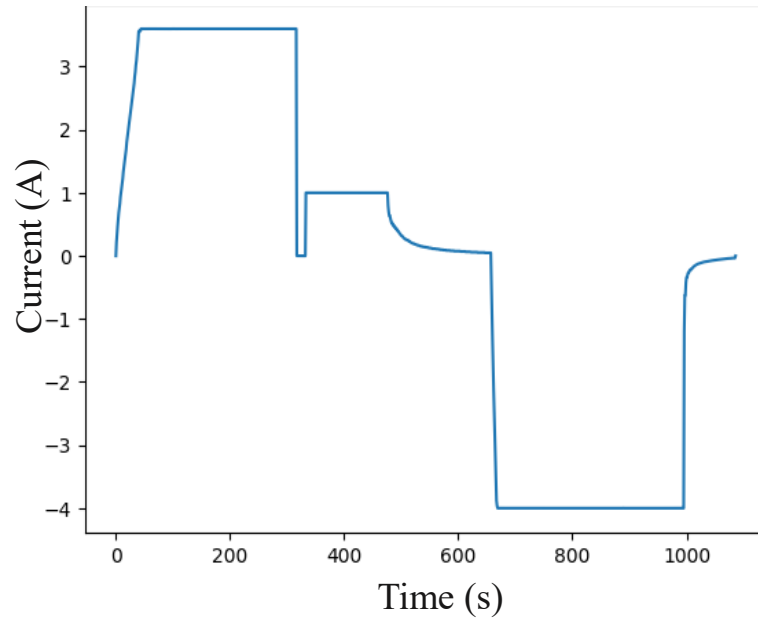


- The combined training and testing data points are 667000

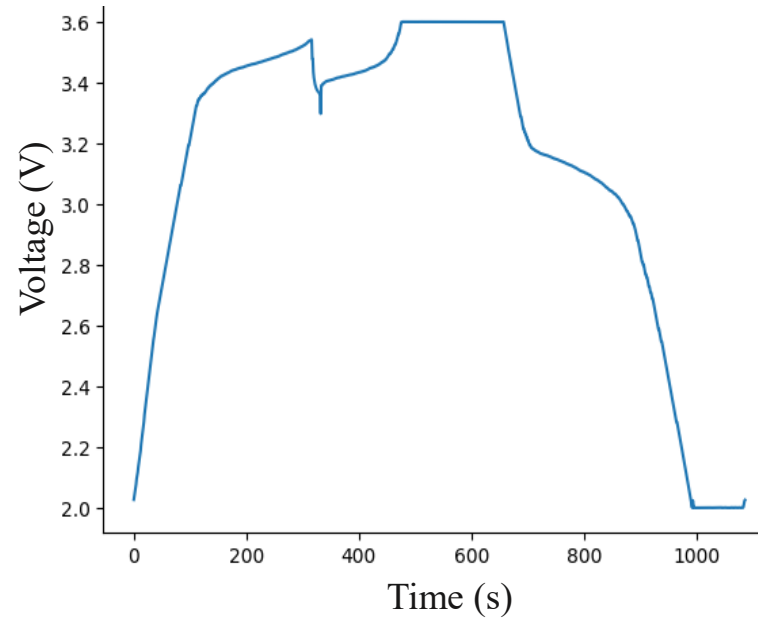


ML Model 1 – Raw Data

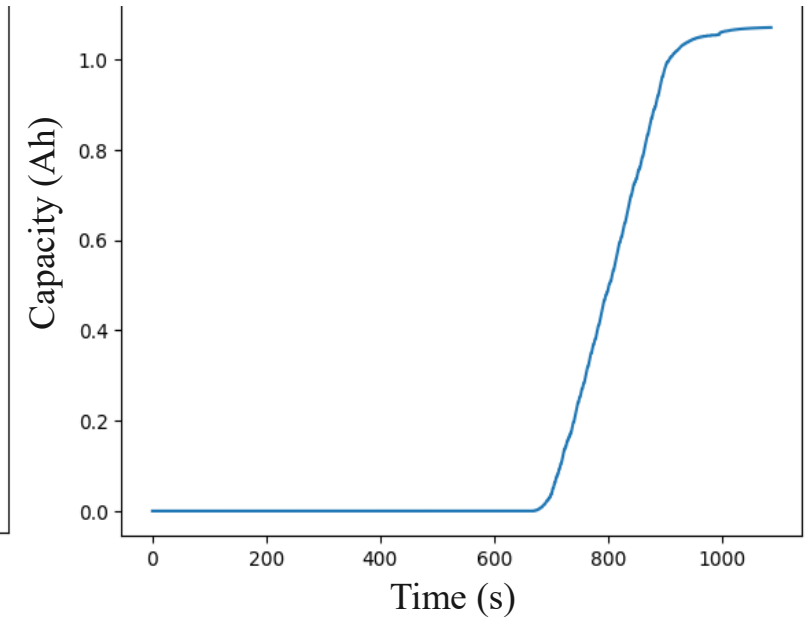
Current Profile for 1 Cycle



Voltage Profile for 1 Cycle

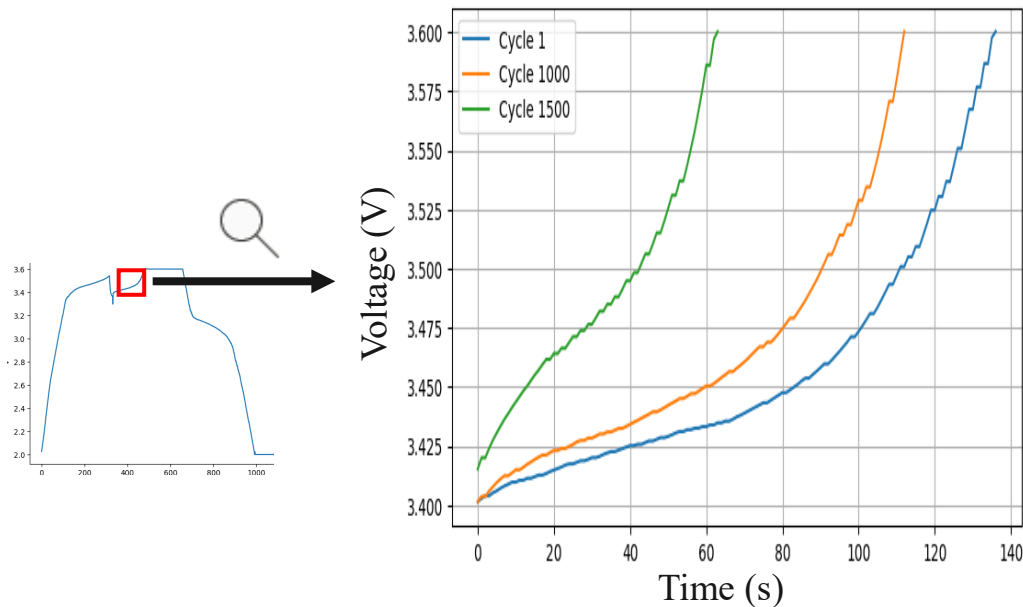


Discharge Capacity Profile for 1 Cycle

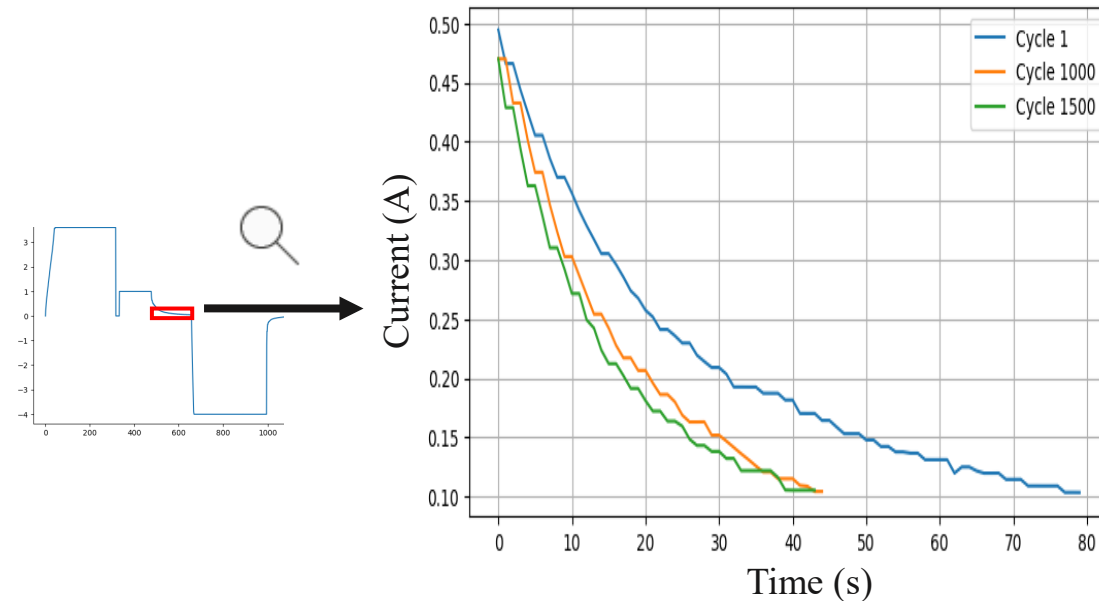


ML Model 1 – Feature Extraction

Voltage Curve



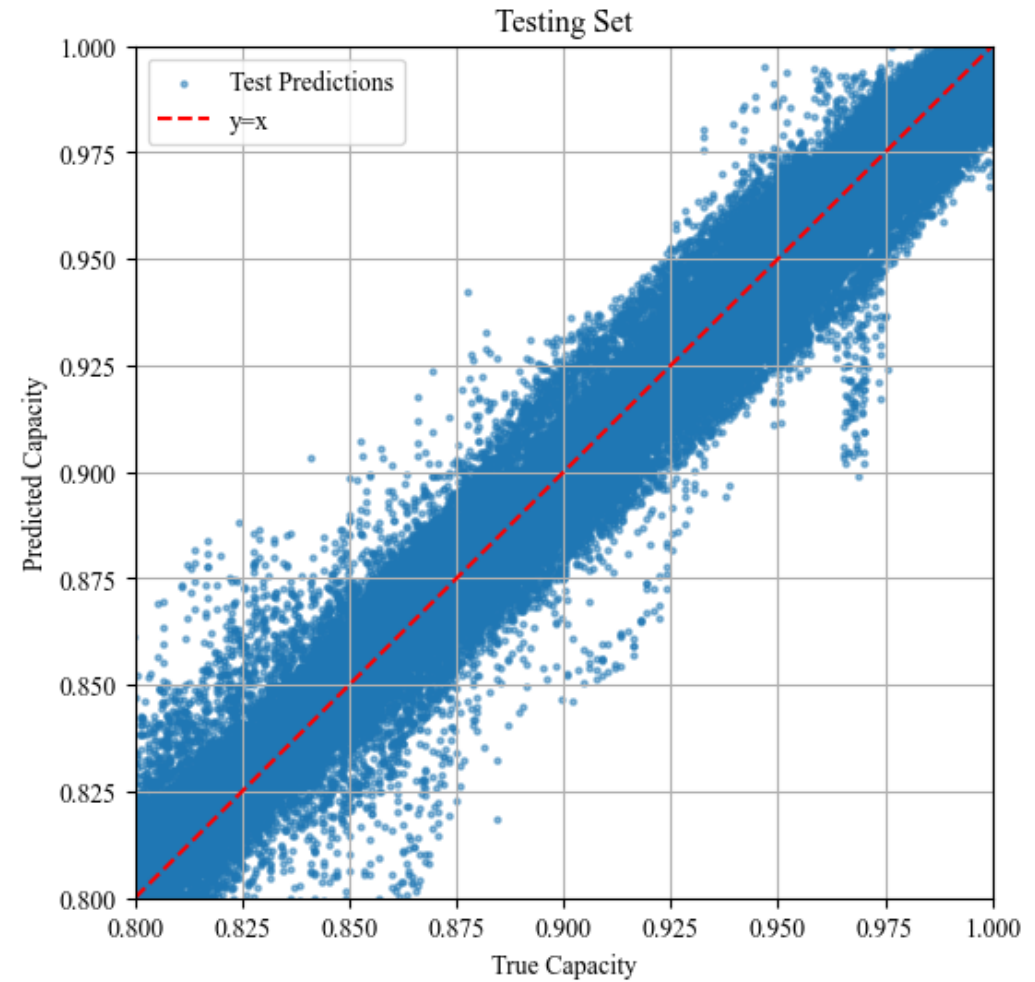
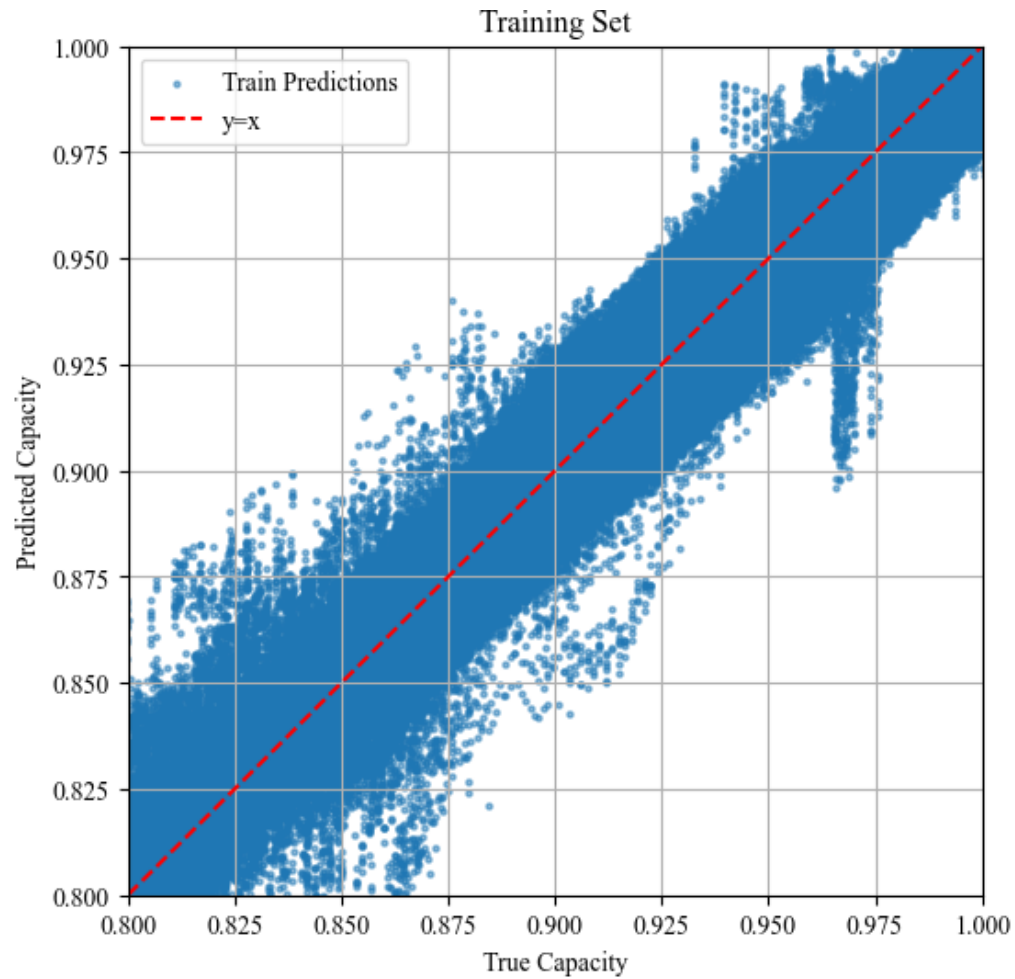
Current Curve



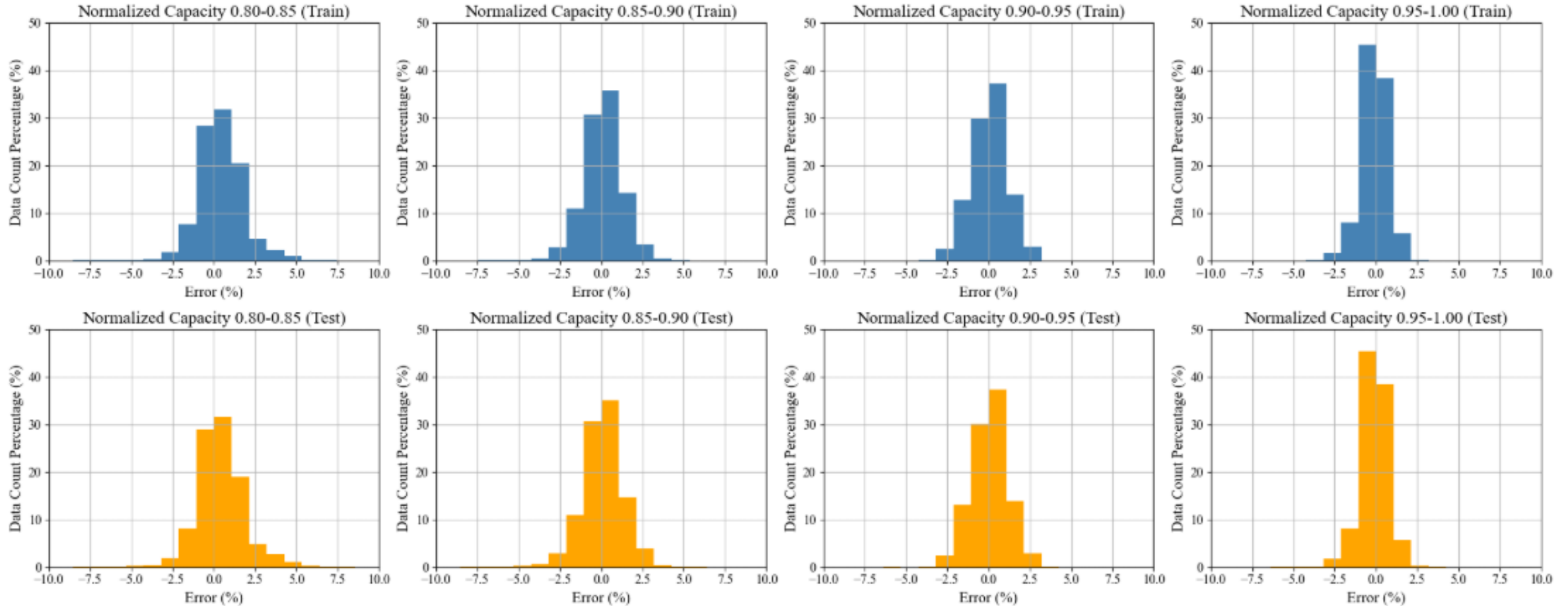
- Features are extracted from the current and voltage curves during a short period before battery is fully charged
- 8 features are extracted from each curve: mean, standard deviation, kurtosis, skewness, duration, accumulated charge, curve slope and curve entropy



ML Model 1 – Modeling Result



ML Model 1 – Modeling Result



- Error is calculated as $(\text{predicted capacity} - \text{true capacity}) / \text{true capacity} \times 100$
- The maximum error for both the training and testing sets is approximately 8.5%



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Current Challenges and Future Work

Challenges

- Due to the lengthy process of conducting battery life cycle tests, it is challenging to gather a dataset with large number of battery cells from either lab or field
- Physics-based models require electrochemical parameters that are often unavailable from field data. Methods must be developed to calibrate these parameters obtained from online sources and the literature

Future Work

- Continue calibrating the ML model 1 of the Physics-informed ML model
- Process newly obtained field data and apply the physics model to it
- Integrate calendar aging into the Physics-informed ML model





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Q & A

THANK YOU!

Appendix - Principle Equations

Doyle-Fuller-Newman Model (DFN)

Lithium Diffusion in Solid Particles

$$\frac{\partial c_s(x, r, t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left(D_s(c_s, T) r^2 \frac{\partial c_s}{\partial r} \right)$$

Lithium Transport in Electrolyte

$$\frac{\partial(\varepsilon_e c_e)}{\partial t} = \nabla \cdot (D_e^{\text{eff}} \nabla c_e) + \frac{1-t^+}{F} a j_{\text{tot}}(x, t)$$

Charge Conservation (electrolyte)

$$\nabla \cdot (\kappa^{\text{eff}} \nabla \phi_e + 2\kappa^{\text{eff}}(1-t^+) \frac{RT}{F} \nabla \ln c_e) = a j_{\text{tot}}(x, t)$$

Charge Conservation (solid)

$$(\sigma^{\text{eff}} \nabla \phi_s) = -a j_{\text{tot}}(x, t)$$

SEI growth

$$\frac{\partial L_{\text{SEI}}(x, t)}{\partial t} = \frac{M_{\text{SEI}}}{\rho_{\text{SEI}} F} j_{\text{SEI}}(x, t)$$

Lithium plating $j_{\text{pl}}(x, t) = i_{0,\text{pl}}(x, t) \left[\exp\left(\frac{\alpha F \eta_{\text{pl}}(x, t)}{RT}\right) - \exp\left(-\frac{(1-\alpha) F \eta_{\text{pl}}(x, t)}{RT}\right) \right]$ $j_{\text{pl}}(t) = i_{0,\text{pl}}(t) \left[\exp\left(\frac{\alpha F \eta_{\text{pl}}(t)}{RT}\right) - \exp\left(-\frac{(1-\alpha) F \eta_{\text{pl}}(t)}{RT}\right) \right]$

Loss of Active Material $j_{\text{int,eff}}(x, t) = f_{\text{LAM}}(x, t) a(x) i_0 \left[e^{\alpha_a F \eta / RT} - e^{-\alpha_c F \eta / RT} \right]$

Single Particle Model (SPM)

$$\frac{\partial c_s(r, t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left(D_s(c_s, T) r^2 \frac{\partial c_s}{\partial r} \right)$$

Not included for SPM

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$$\frac{dL_{\text{SEI}}(t)}{dt} = \frac{M_{\text{SEI}}}{\rho_{\text{SEI}} F} j_{\text{SEI}}(t)$$

